

Electrophoretic deposition of negatively charged tetratitanate nanosheets and transformation into preferentially oriented TiO₂(B) film

Wataru Sugimoto,* Osamu Terabayashi, Yasushi Murakami and Yoshio Takasu

Department of Fine Materials Engineering, Faculty of Textile Science and Technology, Shinshu University, 3-15-1 Tokida, Ueda 386-8567, Japan.

E-mail: wsugi@giptc.shinshu-u.ac.jp

Received 30th April 2002, Accepted 27th August 2002

First published as an Advance Article on the web 18th September 2002

A tetrabutylammonium–H₂Ti₄O₉·xH₂O intercalation compound was obtained by a guest exchange reaction between tetrabutylammonium hydroxide and an ethylammonium–H₂Ti₄O₉·xH₂O intercalation compound, and its dispersion state in aqueous and non-aqueous solutions were studied. Spontaneous exfoliation of H₂Ti₄O₉·xH₂O into colloidal nanosheets occurred when the tetrabutylammonium–H₂Ti₄O₉·xH₂O intercalation compound was dispersed in water, methyl alcohol, isopropyl alcohol, acetonitrile, *N,N*-dimethylformamide, dimethyl sulfoxide, and propylene carbonate, while exfoliation did not occur in tetrahydrofuran.

A tetrabutylammonium–H₂Ti₄O₉·xH₂O film was obtained by a reassembly process by casting the colloidal suspension containing exfoliated nanosheets, while a H₂Ti₄O₉·xH₂O film was directly obtained by electrophoretic deposition. Thermal treatment of the electrophoretically deposited film led to an oriented TiO₂(B) film with the (0*kl*) planes lying perpendicular to the substrate.

Introduction

A variety of layered titanates react with organic species to form intercalation compounds. The reaction of certain alkylamines with layered titanates possessing lepidocrocite-type structure (H_{0.67}Ti_{1.83}□_{0.17}O₄·xH₂O)^{1–18} and Ruddlesden–Popper type structure (H₂La₂Ti₃O₁₀·xH₂O)¹⁹ leads to delamination of the titanate layers into multilamellar or unilamellar nanosheets in a process known as exfoliation. Such nanosheets are promising material for photofunctional material,^{6–8} and have been utilized as building blocks for pillared materials,^{9–12} inorganic–organic composites,^{13,14} and thin films.^{15–19} Thin films using exfoliated nanosheets can be fabricated by spin-coating,^{15–18} or self-assembly on functionalized substrates.¹⁹ The latter method utilizes the electrostatic interaction of the negatively-charged nanosheets for the self-assembly of exfoliated nanosheets.¹⁹

Electrophoretic deposition (EPD) is a well-known technique for obtaining highly uniform films of varying thickness.^{20,21} EPD is based on the electrophoresis of charged particles under the influence of an electric field. Since EPD makes use of the charge of particles for deposition, it is similar to the electrostatic-assembly method. Uniform films from water-swollen clay^{22–25} and exfoliated-cuprate superconductors²⁶ have been fabricated by EPD. The use of EPD for film processing of layered titanates has recently been reported.^{27,28} Kim *et al.* briefly reported on the fabrication of Bi₄Ti₃O₁₂ films by EPD using exfoliated Bi₄Ti₃O₁₂ which was prepared by an “explosive exfoliation” mechanism.²⁷ Matsumoto *et al.* reported that the EPD of H_{0.67}Ti_{1.83}□_{0.17}O₄·xH₂O suspended in ethanol–H₂O affords *b*-axis oriented H_{0.67}Ti_{1.83}□_{0.17}O₄·xH₂O films, while that suspended in *n*-alkylamine–H₂O affords *n*-alkylamine-intercalated H_{0.67}Ti_{1.83}□_{0.17}O₄·xH₂O films.²⁸

In this study, we examined the exfoliation and EPD of a protonated form of layered tetratitanate, H₂Ti₄O₉·xH₂O. H₂Ti₄O₉·xH₂O is the *n* = 4 member of the layered titanate series with the formula H₂Ti_{*n*}O_{2*n*+1}·xH₂O (3 ≤ *n* ≤ 6) and possesses double ReO₃-type chains in which the octahedra

share their edges. The ion-exchange, intercalation, photochemical, and photocatalytic properties of H₂Ti₄O₉·xH₂O have been studied.^{29–47} In addition to such properties, H₂Ti₄O₉·xH₂O is known as the precursor for monoclinic TiO₂, known as TiO₂(B).^{48–54} TiO₂(B) possesses unique electrochemical and catalytic properties compared to other TiO₂ polymorphs.^{55–60} A possible reason for the unique properties of TiO₂(B) is that it possesses a lower density compared to anatase and rutile, which provides cavities and channels for improved insertion and transport of hydrogen or small cations.^{56,61} TiO₂(B) is a metastable form of titanium dioxide which is obtained *via* the so-called “soft chemistry reaction” involving a low-energy transformation of H₂Ti₄O₉·xH₂O into TiO₂(B) upon thermal treatment. Such a distinctive synthetic method has so far been limited to the preparation of TiO₂(B) only in a powder form.

We report here, the exfoliation of H₂Ti₄O₉·xH₂O in various solvents, the fabrication of a H₂Ti₄O₉·xH₂O film by EPD, and its transformation into a TiO₂ film. We have found that electrophoretic deposition affords fabrication of a H₂Ti₄O₉·xH₂O film without an organic cation in the interlayer which allows straightforward conversion into an oriented TiO₂(B) film with the (0*kl*) plane perpendicular to the substrate.

Experimental

The protonated tetratitanate, H₂Ti₄O₉·xH₂O was prepared following a previously reported procedure.³¹ The ethylammonium–H₂Ti₄O₉·xH₂O (EA–H₂Ti₄O₉·xH₂O) intercalation compound was prepared by the reaction of H₂Ti₄O₉·xH₂O with aqueous solutions of *n*-ethylammonium for 24 h at room temperature. EA–H₂Ti₄O₉·xH₂O (dried overnight at 120 °C) was allowed to react with aqueous solutions of tetrabutylammonium hydroxide (TBAOH) at room temperature for 50 h followed by centrifugal collection at 15 000 rpm. The particulate product was subsequently suspended in

various solvents (distilled water, methyl alcohol, isopropyl alcohol, acetonitrile, *N,N*-dimethylformamide, dimethyl sulfoxide, propylene carbonate, and tetrahydrofuran). Relatively large particles and agglomerates were centrifugally separated at 2000 rpm. The supernatant was used for further investigation.

The electrochemical cell for the electrophoretic deposition (EPD) was composed of a Pt plate anode (1 cm²), a Pt plate cathode (6 cm²), and a Ag/AgCl reference electrode. The anode and cathode were placed parallel with a separation of 20 mm. EPD was carried out by applying a constant potential of 10 V (*vs.* Ag/AgCl) for 1 minute at 25 °C. Thermal treatment of the deposited films was conducted at 500 °C for 1 h in air.

The structures of the products were studied by X-ray diffraction (XRD) (Rigaku RINT 2550H/PC; monochromated Cu-K α radiation). Thermal analysis was conducted with a Rigaku TAS-200. Field-emission scanning-electron microscopy (FE-SEM) (Hitachi S-5000) was utilized for morphological observation of the products. In order to observe the morphology of the non-conducting colloidal particles at high magnification without the assistance of metal coatings, which is often used to suppress the effect of charge-up, the colloidal particles were supported on carbon materials (coconut shell activated carbon and Ketjen Black; supplied by Mitsubishi Chemicals).

Results and discussion

The XRD pattern of the product obtained by the reaction of EA-H₂Ti₄O₉·*x*H₂O and aqueous TBAOH is shown in Fig. 1. The basal spacing increased from *d* = 0.98 nm for EA-H₂Ti₄O₉·*x*H₂O to *d* = 1.8 nm. The interlayer distance, *d*_{interlayer}, of the product was estimated using the following equation:

$$d_{\text{interlayer}} = d_{(200)} \times \sin \alpha - d_{\text{oxide sheet}} = 1.2 \text{ nm}$$

where, the values $\alpha = 104^\circ$ and $d_{\text{oxide sheet}} = 0.56 \text{ nm}$ were used based on previous studies.⁴⁷ The interlayer distance of 1.2 nm is approximately equal to the geometric size of the TBA ion,⁴ suggesting a monolayer arrangement in the interlayer. Analysis of the organic content in the product by thermogravimetry indicated a loss of 22.7 mass% which corresponds to the chemical formula TBA_{0.3}H_{1.7}[Ti₄O₉]. These results reveal that ethylammonium in the interlayer

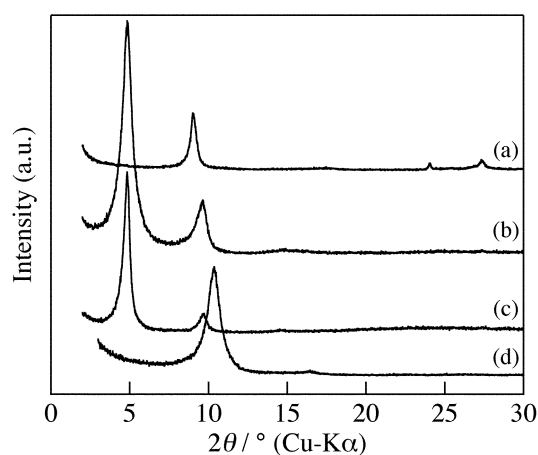


Fig. 1 The XRD patterns of (a) the EA-H₂Ti₄O₉·*x*H₂O intercalation compound, (b) the product obtained by the reaction of aqueous TBAOH with EA-H₂Ti₄O₉·*x*H₂O, (c) a film obtained by casting a colloidal suspension of the product obtained in (b), and (d) a film obtained by electrophoretic deposition of a colloidal suspension of the product obtained in (b).

was exchanged with tetrabutylammonium to form a tetrabutylammonium-H₂Ti₄O₉·*x*H₂O (TBA-H₂Ti₄O₉·*x*H₂O) intercalation compound.

Dispersion of the TBA-H₂Ti₄O₉·*x*H₂O intercalation compound in methyl alcohol resulted in a translucent colloidal suspension that was stable for periods of months. The XRD pattern of a film obtained by casting the colloidal suspension in methyl alcohol onto a glass substrate is shown in Fig. 1c. The XRD pattern of the cast film was identical to that of the TBA-H₂Ti₄O₉·*x*H₂O intercalation compound in powder form, which indicates that casting the colloidal solution results in reassembly of the TBA ions and exfoliated H₂Ti₄O₉·*x*H₂O. Fig. 2 shows the morphology of the exfoliated H₂Ti₄O₉·*x*H₂O supported on various carbon materials observed by FE-SEM. Even with the oxide loading, the morphology of the carbon support could be observed, which shows that the oxide sheets have a high degree of flexibility. The oxide sheets have lateral dimensions in the range of approximately 0.1 × 1 μm and thicknesses in the order of nanometers. The FE-SEM image of montmorillonite, a well-known water swellable clay, is also shown for comparison. The FE-SEM images show that the exfoliated nanosheets have a similar morphology to that of swollen clay.

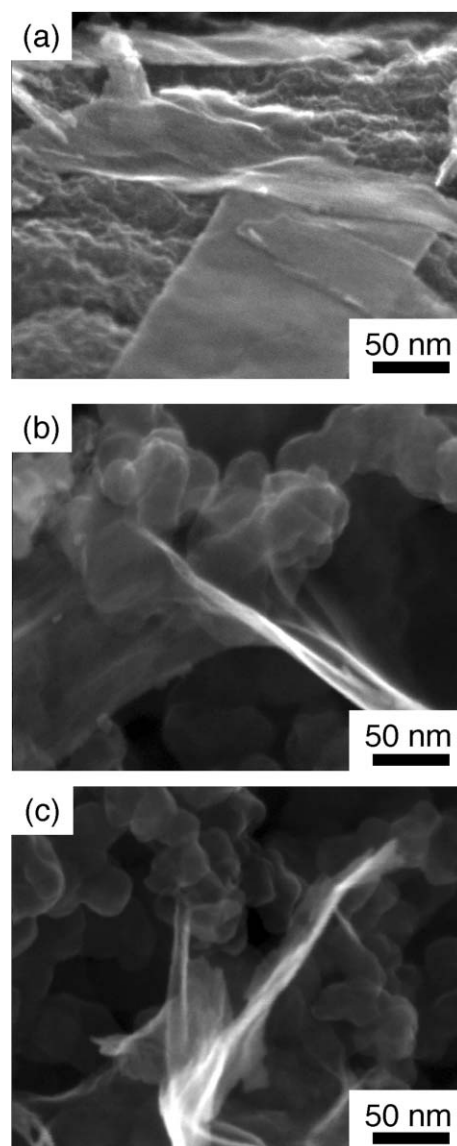


Fig. 2 Typical FE-SEM images of exfoliated H₂Ti₄O₉·*x*H₂O supported on (a) coconut-shell based carbon and (b) Ketjen Black. (c) A typical FE-SEM image of montmorillonite supported on Ketjen Black.

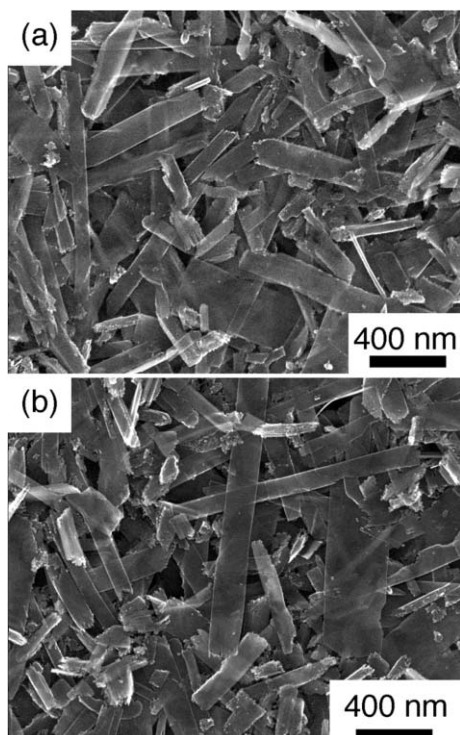


Fig. 3 (a) A typical FE-SEM image of the surface of a film obtained by the EPD from colloidal suspensions of exfoliated $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$. (b) A typical FE-SEM image of a $\text{TiO}_2(\text{B})$ film prepared by 500°C treatment of exfoliated $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$.

Table 1 Summary of the properties of the solvents used for the dispersion of $\text{TBA-H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ ^a

Solvent	Boiling point/ $^\circ\text{C}$	Viscosity ^b / mPa s	Relative permittivity ^b
Water	99.97	0.890	78.30
Methyl alcohol	64.7	0.545	32.7
Isopropyl alcohol	82.3	1.765 ₃₀	19.9
Acetonitrile	81.6	0.325 ₃₀	37.5 ₂₀
Dimethyl sulfoxide	189.0	2.00	46.7
<i>N,N</i> -Dimethylformamide	153.0	0.802	36.7
Propylene carbonate	241.7	2.53	64.4
Tetrahydrofuran	66	0.460	1.75

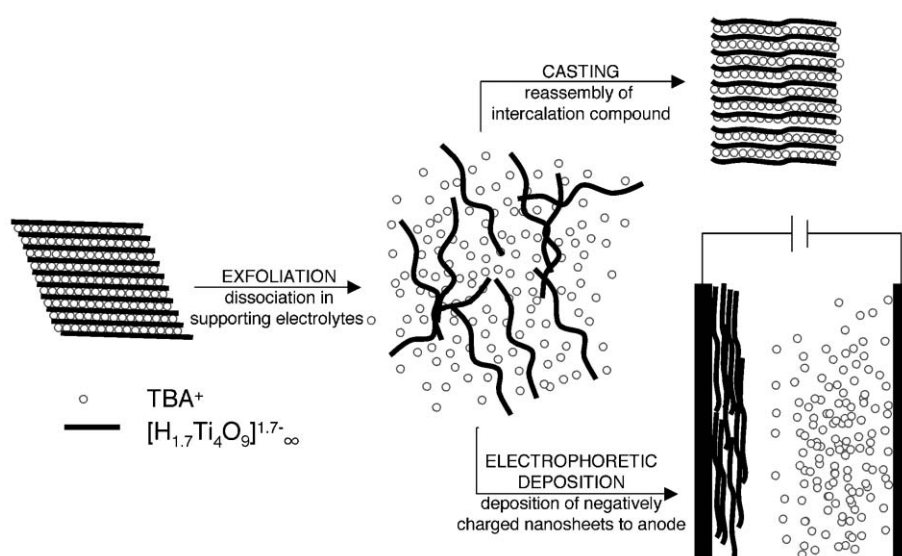
^aData from ref. 62. ^bValues are those at 25°C unless indicated otherwise by the temperatures given as subscripts.

Applying a bias between two electrodes placed in the colloidal suspension of $\text{TBA-H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ in methyl alcohol resulted in deposition at the anode. The film thickness increased as a function of the deposition time; the thickness of the deposited film was 0.6 ± 0.2 , 1.4 ± 0.4 , and $3.0 \pm 0.5 \mu\text{m}$ for deposition times of 0.5, 1, and 3 minutes, respectively.

A typical FE-SEM image of the surface of a film obtained by the EPD of exfoliated $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ is shown in Fig. 3, revealing plate-like sheets uniformly stacked parallel to the substrate. Whereas casting the colloidal suspension onto a glass substrate resulted in a $\text{TBA-H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ intercalation compound (Fig. 1c), the XRD pattern of the EPD film (dried overnight at 120°C) was identified as $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ (Fig. 1d). This suggests that EPD resulted in electrophoresis of the negatively charged exfoliated nanosheets ($[\text{H}_{1.7}\text{Ti}_4\text{O}_9]^{0.3-\infty}$) to the anode and the positively charged TBA ions to the cathode. That is, the $\text{TBA-H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ intercalation compound completely dissociates in methyl alcohol, allowing $\text{TBA-H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ to act as an electrolyte and methyl alcohol to act as the supporting electrolyte. This was further supported by studying the stability of the colloidal suspension in various solvents (distilled water, isopropyl alcohol, acetonitrile, *N,N*-dimethylformamide, dimethyl sulfoxide, propylene carbonate, and tetrahydrofuran). Stable colloidal solutions were obtained for all the electrolytes with the exception of tetrahydrofuran. The relative permittivity of tetrahydrofuran is too low to act as a supporting electrolyte for the dissociation of negatively-charged titanate sheets and positively-charged TBA ions (Table 1). A proposed mechanism is shown in Scheme 1.

These results differ from that of the EPD of $\text{H}_{0.67}\text{Ti}_{1.83}\square_{0.17}\text{O}_4 \cdot x\text{H}_2\text{O}$ suspended in aqueous *n*-alkylamine which resulted in the deposition of the *n*-alkylammonium- $\text{H}_{0.67}\text{Ti}_{1.83}\square_{0.17}\text{O}_4 \cdot x\text{H}_2\text{O}$ intercalation compound.²⁸ Domen *et al.* have reported that the reaction of $\text{H}_{0.67}\text{Ti}_{1.83}\square_{0.17}\text{O}_4 \cdot x\text{H}_2\text{O}$ with aqueous ethylamine solution does not lead to delamination into unilamellar sheets, but partial delamination into 10–15 layer multilamellar sheets.¹⁷ Hence, the EPD of $\text{H}_{0.67}\text{Ti}_{1.83}\square_{0.17}\text{O}_4 \cdot x\text{H}_2\text{O}$ suspended in aqueous *n*-alkylamine probably led to the deposition of partially delaminated, multilamellar *n*-alkylamine- $\text{H}_{0.67}\text{Ti}_{1.83}\square_{0.17}\text{O}_4 \cdot x\text{H}_2\text{O}$ intercalation compound. Bulky cations such as tetrabutylammonium are probably favorable for the complete delamination to unilamellar sheets.

The XRD pattern of the EPD film after thermal treatment at 500°C is shown in Fig. 4. The XRD pattern closely



Scheme 1 Schematic of the exfoliation and deposition of $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$.

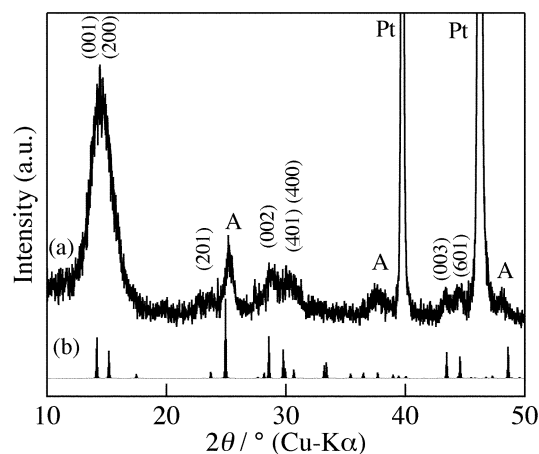


Fig. 4 (a) The XRD pattern of electrophoretically deposited film after thermal treatment at 500 °C and (b) the simulated XRD pattern of $\text{TiO}_2(\text{B})$ using the parameters reported by Fiest and Davies (ref. 53). Anatase impurities are marked with an A.

matched that of $\text{TiO}_2(\text{B})$, except for the fact that only the $(h0l)$ reflections were observed; that is, (110) , (310) , $(\bar{3}11)$, (020) were not observed at $2\theta = 25.0, 33.2, 33.4, 48.6^\circ$. Some weak reflections due to anatase were also observed, but this impurity is very difficult to avoid even by careful optimization of the conditions for thermal treatment.⁵³ The absence of (hkl) reflections with $k \neq 0$ indicates that $\text{TiO}_2(\text{B})$ is oriented with the $(0k0)$ planes lying perpendicular to the substrate. As shown in Fig. 3b, the plate-like morphology of the as-deposited films was preserved in the film treated at 500 °C. Hence, the preferred orientation of $\text{TiO}_2(\text{B})$ must be related to the a -axis orientation of the as-deposited $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ film.

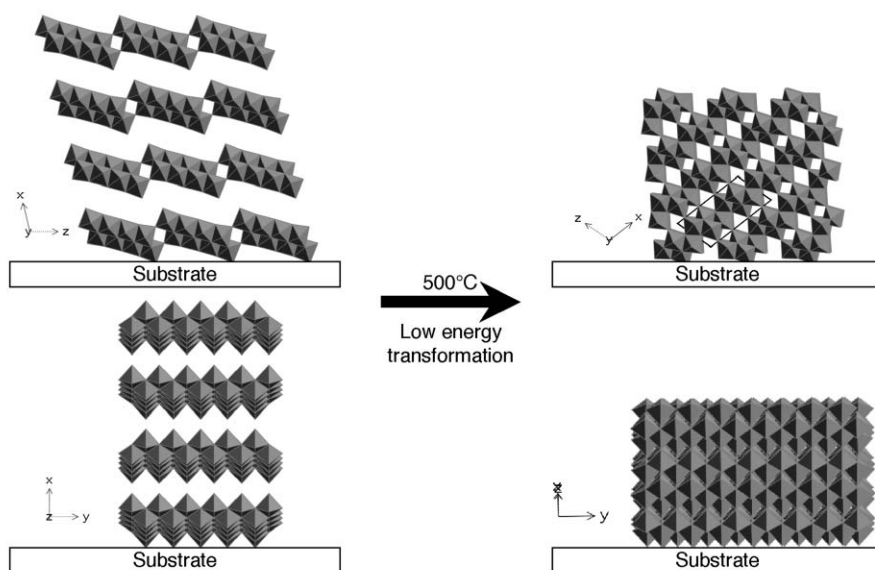
The transformation of $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ to $\text{TiO}_2(\text{B})$ has been thoroughly discussed by Fiest and Davis.⁵³ The transformation is not a simple condensation of the interlayer hydroxy groups, and involves a low energy transformation with several transition steps. Based on the above results and the structural similarity between $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ and $\text{TiO}_2(\text{B})$, it can be said that a (201) -oriented $\text{TiO}_2(\text{B})$ film was derived from an a -axis oriented $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ film as shown schematically in Scheme 2. The present results indicate that the transformation of $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ into $\text{TiO}_2(\text{B})$ is of low enough energy that

the morphology and preferred orientation of the precursor layered tetratitanate is reflected in the resulting $\text{TiO}_2(\text{B})$ film. Thermal treatment at 700 °C leads to a breakdown of the sheets into 20–30 nm sized anatase particles.

Finally, it should be noted that $\text{TiO}_2(\text{B})$ was not observed in the XRD patterns when a cast film using exfoliated $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ in aqueous TBAOH or an electrophoretically deposited film using exfoliated $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ in propylene carbonate was thermally treated at 500 °C. In these films, XRD revealed that TBA or propylene carbonate was present in the interlayer. The presence of such organic compounds in the interlayer seems to hinder the transformation of the two-dimensional structure into the three-dimensional tunnel structure, which has been proposed to be the intermediate phase.⁵³ The application of EPD for film processing using exfoliated $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ presents a significant advantage in thin film preparation of metastable oxides such as $\text{TiO}_2(\text{B})$ since such oxides cannot be obtained by other synthetic methods (chemical vapor deposition, sol-gel, etc.).

Conclusions

An oriented $\text{TiO}_2(\text{B})$ film was prepared by thermal treatment of an a -axis oriented $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ film fabricated by electrophoretic deposition using colloidal suspensions containing exfoliated $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$. Stable colloidal suspensions containing exfoliated $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ were prepared by dispersing a tetrabutylammonium- $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ intercalation compound in various solvents. It was proposed that the positively charged tetrabutylammonium cation and negatively charged exfoliated nanosheets completely dissociate in solvents with high relative permittivity. Electrophoretic deposition of the colloidal suspensions lead to an a -axis oriented $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ film. The a -axis oriented $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ film was converted into a $\text{TiO}_2(\text{B})$ film with the $(0k0)$ planes lying perpendicular to the substrate by thermal treatment. Since bulk $\text{TiO}_2(\text{B})$ is known to possess unique electrochemical and catalytic properties compared to other TiO_2 polymorphs, the properties of the $\text{TiO}_2(\text{B})$ films prepared in this study are of considerable interest. Furthermore, the application of electrophoretic deposition to fabricate thin films from exfoliated oxide suspensions opens up the possibility of preparing novel metastable oxide films.



Scheme 2 Schematic showing the transformation of (left) a (200) -oriented $\text{H}_2\text{Ti}_4\text{O}_9 \cdot x\text{H}_2\text{O}$ film to (right) a (201) -oriented $\text{TiO}_2(\text{B})$.

References

- 1 T. Sasaki, M. Watanabe, H. Hashizume, H. Yamada and H. Nakazawa, *J. Am. Chem. Soc.*, 1996, **118**, 8329–8335.
- 2 T. Sasaki, M. Watanabe, H. Hashizume, H. Yamada and H. Nakazawa, *Chem. Commun.*, 1996, 229–230.
- 3 T. Sasaki and M. Watanabe, *J. Am. Chem. Soc.*, 1998, **120**, 4682–4689.
- 4 T. Sasaki and M. Watanabe, *Mol. Cryst. Liq. Cryst.*, 1998, **11**, 417–422.
- 5 T. Sasaki, *Supramol. Sci.*, 1998, **5**, 367–371.
- 6 T. Sasaki and M. Watanabe, *J. Phys. Chem.*, 1997, **101**, 10159–10161.
- 7 T. Sasaki, S. Nakano, S. Yamauchi and M. Watanabe, *Chem. Mater.*, 1997, **9**, 602–608.
- 8 T. Sasaki, Y. Ebina, M. Watanabe and G. Decher, *Chem. Commun.*, 2000, 2163–2164.
- 9 F. Kooli, T. Sasaki and M. Watanabe, *Microporous Mesoporous Mater.*, 1999, **28**, 495–503.
- 10 F. Kooli, T. Sasaki and M. Watanabe, *Chem. Commun.*, 1999, 211–212.
- 11 F. Kooli, T. Sasaki, V. Rives and M. Watanabe, *J. Mater. Chem.*, 2000, **10**, 497–501.
- 12 J. H. Choy, H. C. Lee, H. Jung and S. J. Hwang, *J. Mater. Chem.*, 2001, **11**, 2232–2234.
- 13 T. Sasaki, Y. Ebina, Y. Kitami, M. Watanabe and T. Oikawa, *J. Phys. Chem. B*, 2001, **105**, 6116–6121.
- 14 N. Sukpirom and M. L. Lerner, *Chem. Mater.*, 2001, **13**, 2179–2185.
- 15 R. Abe, K. Shinohara, A. Tanaka, M. Hara, J. N. Kondo and K. Domen, *Chem. Mater.*, 1998, **10**, 329–333.
- 16 R. Abe, S. Ikeda, J. N. Kondo, M. Hara and K. Domen, *Thin Solid Films*, 1999, **343–344**, 156–159.
- 17 T. Sumida, R. Abe, M. Hara, J. N. Kondo and K. Domen, *Electrochemistry*, 1999, **67**, 1224–1226.
- 18 M. Koinuma, Y. Matsumoto, T. Sumida and K. Domen, *Electrochem. Solid State Lett.*, 2000, **3**, 481–484.
- 19 R. E. Schaak and T. E. Mallouk, *Chem. Mater.*, 2000, **12**, 3427–3434.
- 20 P. Sarkar and P. S. Nicholson, *J. Am. Ceram. Soc.*, 1996, **79**, 1987–2002.
- 21 M. S. J. Gani, *Indust. Ceram.*, 1994, **14**, 163–174.
- 22 C. Hirayama and D. Berg, *Electrochem. Technol.*, 1963, **1**, 224–227.
- 23 J. S. Graham and D. R. Rosseinsky, *J. Chem. Soc., Chem. Commun.*, 1994, 2389–2390.
- 24 J. S. Graham, J. D. Rosseinsky, J. D. Slocombe, S. Barrett and S. R. Francis, *Colloids Surf. A*, 1995, **94**, 177–188.
- 25 C. Song and G. Villemure, *J. Electroanal. Chem.*, 1999, **462**, 143–149.
- 26 J. H. Choy, S. J. Kwon, S. J. Hwang and E. S. Jang, *MRS Bull.*, 2000, **25**, 32–39.
- 27 J. Y. Kim, I. Chung and J. H. Choy, *Chem. Mater.*, 2001, **13**, 2759–2761 (Supplementary Information).
- 28 Y. Matsumoto, A. Funatsu, D. Matsuo, U. Unal and K. Ozawa, *J. Phys. Chem. B*, 2001, **105**, 10893–10899.
- 29 M. Dion, Y. Piffard and M. Tournoux, *J. Inorg. Nucl. Chem.*, 1978, **40**, 917–918.
- 30 B. Raveau, *Rev. Inorg. Chem.*, 1987, **9**, 37–64.
- 31 H. Izawa, S. Kikkawa and M. Koizumi, *J. Phys. Chem.*, 1982, **86**, 5023–5026.
- 32 H. Izawa, S. Kikkawa and M. Koizumi, *Polyhedron*, 1983, **2**, 741–744.
- 33 S. Cheng and T. C. Wang, *Inorg. Chem.*, 1989, **28**, 1283–1289.
- 34 T. Sasaki, Y. Komatsu and Y. Fujiki, *Inorg. Chem.*, 1989, **28**, 2776–2779.
- 35 T. Sato, K. Masaki, Y. Yoshioka and A. Okuwaki, *J. Chem. Tech. Biotechnol.*, 1993, **58**, 315–319.
- 36 W. Hou, Q. Yan and X. Fu, *J. Chem. Soc., Chem. Commun.*, 1994, 1371–1372.
- 37 W. H. Hou, Q. Yan, B. Peng and X. Fu, *J. Mater. Chem.*, 1995, **5**, 109–111.
- 38 J. H. Choy, Y. S. Han, N. G. Park, H. Kim and S. W. Kim, *Synth. Met.*, 1995, **71**, 2053–2054.
- 39 T. Sato, K. Masaki, K. Sato, Y. Fujishiro and A. Okuwaki, *J. Chem. Tech. Biotechnol.*, 1996, **67**, 339–344.
- 40 S. Uchida, Y. Yamamoto, Y. Fujishiro, A. Watanabe, O. Ito and T. Sato, *J. Chem. Soc., Faraday Trans.*, 1997, **93**, 3229–3234.
- 41 S. Zheng, D. Yin, W. Miao and G. K. Anderson, *J. Photochem. Photobiol., A*, 1998, **117**, 105–109.
- 42 W. Hou, Y. Chen, C. Guo and Q. Yan, *J. Solid State Chem.*, 1998, **136**, 320–321.
- 43 Y. Komatsu, S. Umetani, S. Tsurubou, Y. Michiue and T. Sasaki, *Chem. Lett.*, 1998, 1167–1168.
- 44 M. Ogawa and Y. Takizawa, *Chem. Mater.*, 1999, **11**, 30–32.
- 45 S. Yin and T. Sato, *Ind. Eng. Chem. Res.*, 2000, **39**, 4526–4530.
- 46 S. Ogura, K. Sato and Y. Inoue, *Phys. Chem. Chem. Phys.*, 2000, **2**, 2449–2454.
- 47 M. Yanagisawa, S. Uchida, S. Yin and T. Sato, *Chem. Mater.*, 2001, **13**, 174–178.
- 48 R. Marchand, L. Brohan and M. Tournoux, *Mater. Res. Bull.*, 1980, **15**, 1129–1133.
- 49 L. Brohan, A. Verbaere, M. Tournoux and G. Demazeau, *Mater. Res. Bull.*, 1982, **17**, 355–361.
- 50 R. Marchand, L. Brohan, R. M'Bedi and M. Tournoux, *Rev. Chim. Miner.*, 1984, **21**, 476–486.
- 51 M. Tournoux, R. Marchand and L. Brohan, *Prog. Solid State Chem.*, 1986, **17**, 33–52.
- 52 T. P. Fiest, S. J. Mocariski, P. K. Davies, A. J. Jacobson and J. T. Lewandowski, *Solid State Ionics*, 1988, **28–30**, 1338–1343.
- 53 T. P. Fiest and P. K. Davies, *J. Solid State Chem.*, 1992, **101**, 275–295.
- 54 S. Yin, S. Uchida, Y. Fujishiro, M. Aki and T. Sato, *J. Mater. Chem.*, 1999, **9**, 1191–1195.
- 55 L. Brohan and R. Marchand, *Solid State Ionics*, 1983, **9&10**, 419–424.
- 56 G. Betz, H. Tributsch and R. Marchand, *J. Appl. Electrochem.*, 1984, **14**, 315–322.
- 57 J. Papachryssanthou, E. Bordes, A. Vejux, P. Courtine, R. Marchand and T. Tournoux, *Catal. Today*, 1987, **1**, 219–227.
- 58 M. Sanati and A. Andersson, *J. Mol. Catal.*, 1990, **59**, 233–255.
- 59 S. Jansen, Y. Tu, M. J. Palmieri, M. Sanati and A. Andersson, *J. Catal.*, 1992, **138**, 79–89.
- 60 M. Sanati, R. Akbari, S. Masetti and F. Trifirò, *Catal. Today*, 1998, **42**, 325–332.
- 61 D. W. Murphy and P. A. Christian, *Science*, 1979, **205**, 651–656.
- 62 J. A. Riddick and W. B. Bunger, *Organic Solvents*, 3rd edn., Wiley-Interscience, New York, 1970.